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Annealing effects on the optical and morphological properties of ZnO nanorods on AZO substrate by using aqueous solution method at low temperature

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Abstract

Vertically aligned ZnO nanorods (NRs) on aluminum-doped zinc oxide (AZO) substrates were fabricated by a single-step aqueous solution method at low temperature. In order to optimize optical quality, the effects of annealing on optical and structural properties were investigated by scanning electron microscopy, X-ray diffraction, photoluminescence (PL), and Raman spectroscopy. We found that the annealing temperature strongly affects both the near-band-edge (NBE) and visible (defect-related) emissions. The best characteristics have been obtained by employing annealing at 400°C in air for 2 h, bringing about a sharp and intense NBE emission. The defect-related recombinations were also suppressed effectively. However, the enhancement decreases with higher annealing temperature and prolonged annealing. PL study indicates that the NBE emission is dominated by radiative recombination associated with hydrogen donors. Thus, the enhancement of NBE is due to the activation of radiative recombinations associated with hydrogen donors. On the other hand, the reduction of visible emission is mainly attributed to the annihilation of OH groups. Our results provide insight to comprehend annealing effects and an effective way to improve optical properties of low-temperature-grown ZnO NRs for future facile device applications.

Keywords: Zinc oxide; Photoluminescence; Raman; Annealing

Background

ZnO is a promising II-VI compound semiconductor because of its excellent catalytic, optoelectronic, and piezoelectric properties. It has been demonstrated to have diverse applications in electronic, optoelectronic, and electrochemical devices, such as ultraviolet (UV) lasers, light-emitting diodes, high-performance nanosensors, and solar cells [1-6]. In addition to the low cost, ease of availability, and chemical stability, the wide direct bandgap of 3.37 eV and large excitonic binding energy (60 meV at 300 K) make ZnO a highly competitive material to GaN. It was also reported that textured ZnO films may have higher quantum efficiency than GaN films [7]. Nowadays, ZnO thin films and nanostructures can be synthesized by using

various deposition techniques, such as molecular beam epitaxy (MBE) [8], pulsed laser deposition (PLD) [9], metal-organic chemical vapor deposition (MOCVD) [10], chemical vapor deposition (CVD) [11-13], and aqueous solution deposition [14,15].

High-temperature techniques such as CVD and thermal evaporation have been mainly employed to grow aligned ZnO nanostructures, for example, nanorods (NRs). These processes have disadvantages of high energy consumption and requirement of expensive infrastructure. Here, we adopt an inexpensive and simple method to prepare uniformly distributed and well-aligned vertical ZnO NRs, whereas no catalyst or seeding step is required to initiate controlled growth. This approach is based on a one-step electrochemical processing of reliably nontoxic and abundant materials in aqueous solution at low temperature ($\leq 80^\circ\text{C}$). Moreover, it allows for large-scale processing at low cost and facile integration for complex devices. The substrate of our choice is aluminum-doped zinc oxide

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(AZO). Meanwhile, transparent and conductive AZO substrate is an alternative to the ITO glass. AZO has better lattice matching than ZnO, so the development of ZnO/AZO devices is now a hot pursuit.

It is known that due to oxygen vacancies (V_O), inherent n-type ZnO is formed and its carrier concentration depends on post-growth annealing treatment. The intrinsic defects in ZnO are always associated with various deposition processes. The understanding of defect properties is very useful to improve the quality of ZnO. Generally, post-deposition annealing treatment is a convenient and appropriate way to modify intrinsic defects and improve the crystallinity of ZnO. Proper annealing is an effective way to obtain high-quality ZnO material. There are many reports on the thermal treatment of ZnO for different annealing conditions such as annealing temperatures and gas environments to improve the optical properties of ZnO. In this paper, ZnO NRs were synthesized by an aqueous solution deposition method and effects of post-growth annealing were studied. The structural, morphological, and optical characteristics have been studied after annealing processes. Mechanisms that are responsible for the annealing effects are investigated.

Methods

ZnO NRs were deposited on AZO substrates by an aqueous solution method. Zinc nitrate hexahydrate (Alfa Aesar, Ward Hill, MA, USA) was used as the zinc source. Ethanol amine (Merck, Whitehouse Station, NJ, USA) and hexamethylenetetramine (HMTA) were used as the stabilizer and base, respectively. Firstly, the AZO substrates were cleaned through sonication in a mixture of acetone and isopropyl alcohol (1:1), followed by cleaning with deionized water and drying in N_2 atmosphere before use. The zinc precursor solution was prepared by dissolving equimolar zinc nitrate hexahydrate, HMTA, and ethanol amine in deionized water. The above solution was stirred by using a magnetic stirrer at 60°C for 10 min. NRs were grown by dipping the as-cleaned substrate horizontally into the prepared solution and were covered with a lid for 30 min at 80°C on a regular laboratory hot plate. The prepared NRs were annealed in air for 2 h by using a microprocessor-controlled furnace for different annealing temperatures ranging from 200°C to 600°C . X-ray diffraction (XRD) was conducted to examine the structure and orientation of ZnO. The surface morphology of the prepared NRs was investigated by scanning electron microscopy (SEM) (JEOL 6380, JEOL Ltd., Akishima-shi, Japan). For the photoluminescence (PL) investigations, the samples were excited by a chopped He-Cd laser beam working at 325 nm. The PL signal was dispersed by a Jobin Yvon Triax 550 monochromator (Jobin Yvon Inc., Edison, NJ, USA) equipped with a 2,400 rules/mm grating. A Hamamatsu R928 photomultiplier tube (Hamamatsu Photonics K.K., Iwata, Japan)

equipped with a lock-in amplifier was used to record the optical intensity of the selected emission. A closed-cycle optical cryostat was used for low temperatures down to 10 K. Room-temperature (RT) Raman scattering measurements were performed in a backscattering configuration on a micro-Raman setup equipped with a Jobin Yvon iHR320 spectrometer and a multi-channel TE-cooled (-70°C) CCD detector.

Results and discussion

Evaluation of the as-grown sample

Figure 1a shows the tilt-view SEM image of the as-grown ZnO NRs on an AZO substrate. A high density of ZnO NRs grew vertically on the substrate. The diameter of the nanorods is about 200 nm. The crystallinity of the grown ZnO NRs was investigated by using XRD. As shown in Figure 1b, the XRD pattern of θ - 2θ scan of the as-grown ZnO NRs shows only the ZnO (002) peak (black solid curve), indicating that the *c*-plane of ZnO is oriented parallel to the basal plane of the AZO substrate. It indicates that individual ZnO NRs, crystallized along

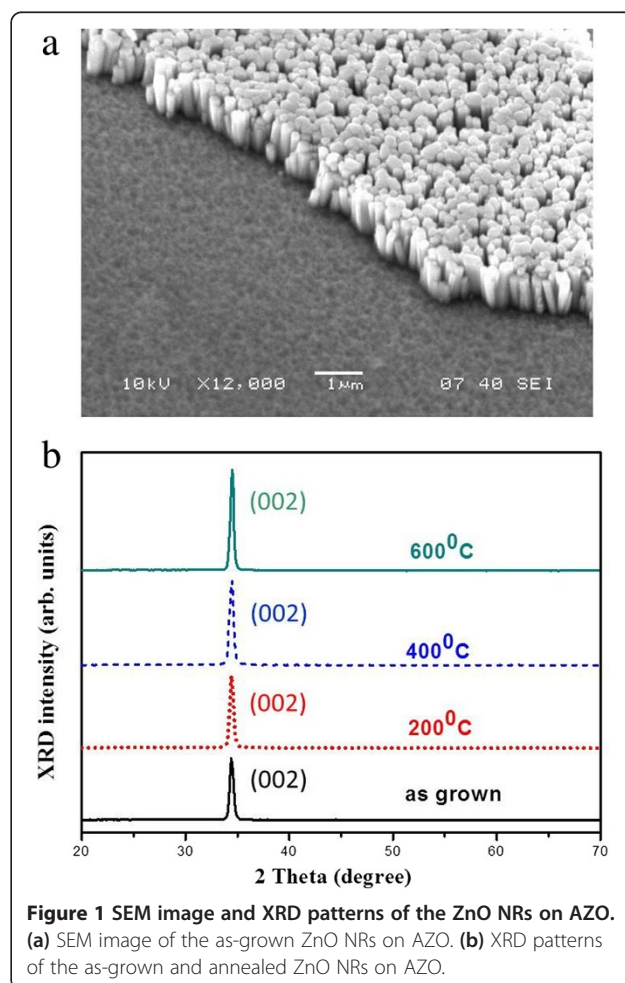


Figure 1 SEM image and XRD patterns of the ZnO NRs on AZO. **(a)** SEM image of the as-grown ZnO NRs on AZO. **(b)** XRD patterns of the as-grown and annealed ZnO NRs on AZO.

the *c*-axis direction of ZnO, were all vertically aligned on the AZO substrate.

The black solid curve in Figure 2 shows the full RT PL spectrum of the as-prepared ZnO NRs. The PL spectrum comprises one UV emission band with a peak at 3.28 eV, which is attributed to the near-band-edge (NBE) emission. In addition, there is a broad visible emission band with comparable intensity, centered at approximately 2.15 eV, which can be ascribed to the defect emission (DE) [16,17]. It is frequently observed in ZnO prepared by aqueous solution and is often considered to be caused by atomic defects, such as oxygen vacancies in ZnO. It means that a lot of photo-generated carriers in ZnO NRs do not recombine close to the band edge, resulting in a poor NBE efficiency.

Effects of annealing on structural and optical properties

In order to improve the optical properties, we performed post-deposition thermal treatment in air at different temperatures for 2 h. Figure 1b displays XRD patterns of the annealed ZnO NRs. The intensities of the (002) diffractions are relatively higher for the samples annealed at various temperatures, suggesting the improvement of the crystalline quality. Then, the effects of thermal annealing on the PL properties of ZnO NRs were investigated. As Figure 2 shows, we found that there is pronounced influence on the NBE emission after annealing. The inset of Figure 2 shows the intensity ratio of NBE to visible DE against annealing temperature. It can be found that the ratio is 1, 4.38, 6, and 1.45 for the as-grown and 200°C-, 400°C-, and 600°C-annealed samples, respectively. We get the strongest NBE from the sample annealed at 400°C. However, the NBE decreases again with the high temperature treatment at 600°C. Moreover, it was noted that the annealing time of 2 h is the optimized duration.

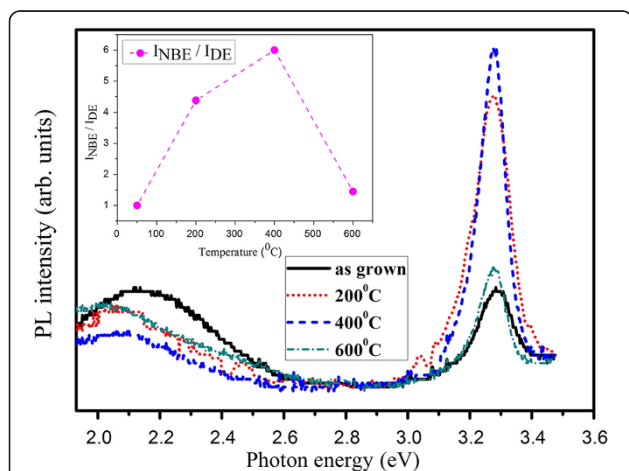


Figure 2 The full spectra of RT PL of the as-grown and annealed ZnO NRs on AZO. The inset shows the NBE to visible DE ratio against annealing temperature.

Figure 3 presents the RT PL spectra of the samples annealed at 400°C in different duration time ranging from 60 to 150 min. It shows that the intensity of NBE is enhanced with increasing annealing time until 120 min. However, it is weakened with prolonged annealing time.

To understand the influence of annealing on the morphological properties, we performed SEM measurements. Figure 4 shows the SEM images of the as-grown ZnO NRs and annealed ZnO NRs at various annealing temperatures. It can be observed that the hexagonal crystallite appears in all the samples, having average diameter ranging from 200 to 300 nm. The whole surface looks smooth and uniform in the nonannealed and annealed (at 200°C and 400°C) samples. However, at 600°C annealing temperature, surface smoothness and uniformity reduce dramatically and some void space is presented on the surface. It suggests that with increasing annealing temperatures, small crystallites start to coalesce together to form larger crystallites [18]. It may be attributed to the annealing-induced coalescence of small grains by grain boundary diffusion [18]. SEM images also indicate that hexagonal crystal phase is less distinct for the as-grown sample. Distinct hexagonal phase appears gradually with increasing annealing temperatures. Furthermore, it appears that the sample with 400°C annealing is best with respect to the homogeneous crystallinity. Together with previous PL results, it implies that 400°C is the optimum annealing temperature to get high-quality ZnO NRs.

Low-temperature PL characterization

To reveal more optical properties, the 10 K PL spectra for the as-grown and 400°C-annealed samples are shown in Figure 5. It is clear that the NBE for the annealed

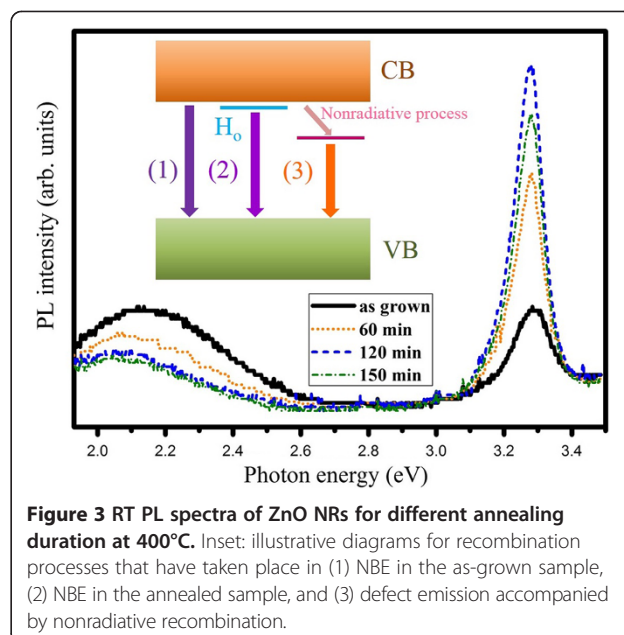


Figure 3 RT PL spectra of ZnO NRs for different annealing duration at 400°C. Inset: illustrative diagrams for recombination processes that have taken place in (1) NBE in the as-grown sample, (2) NBE in the annealed sample, and (3) defect emission accompanied by nonradiative recombination.

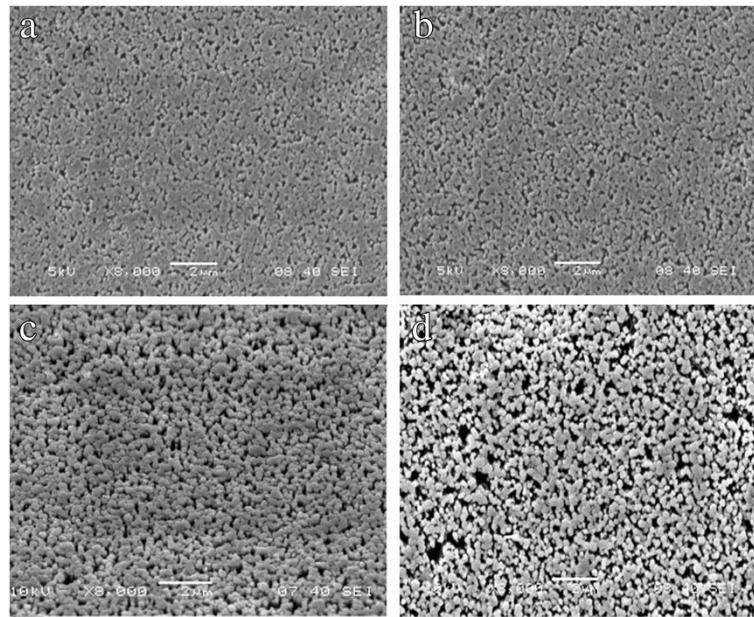


Figure 4 SEM images. (a) As-grown ZnO. Annealed ZnO in air at the following temperatures: (b) 200°C, (c) 400°C, and (d) 600°C.

sample is stronger and sharper than that for the as-grown sample. The full width at half maximum is 47 and 23.5 meV for the as-grown and 400°C-annealed samples, respectively. It gives a quantitative measure of the improved optical quality of our annealed sample. Moreover, we find that there are 3-meV redshifts of NBE emission after annealing at 400°C, as shown by the vertical dashed lines in Figure 5. In order to understand the nature of the enhanced NBE emission, we check the temperature-dependent activated behavior in the inset of

Figure 5. In general, the emission efficiency is determined by the competition of radiative and nonradiative mechanisms. The radiative transition is assumed independent of temperature. On the other hand, the non-radiative contribution has a temperature dependence of $\exp(-E_a/k_B T)$, where E_a is the activation energy and k_B is the Boltzmann constant [19]. In this case, it leads to an expression described by the Arrhenius equation:

$$I(T) = \frac{I_0}{1 + P \exp(-E_a/k_B T)}, \quad (1)$$

where $I(T)$ is the integrated PL intensity at T (K), I_0 is a scaling factor, and P is a process rate parameter [19-21]. It provides important information to the origin of carrier recombination in various semiconductors [21-23]. The dashed line in the inset of Figure 5 is the least square fit of data with Equation 1. The fitted value of P and E_a is 5.2 and 19.5 meV, respectively. We obtain a high thermal activation energy. It is in close agreement with the activation energy of the hydrogen donor in ZnO deduced by PL (21 to 25 meV) [23,24]. Therefore, it is indicative that excitonic recombination at the hydrogen donor (H_O) dominates the NBE after annealing treatment. Next, the temperature dependence of the emission peak $E(T)$ was studied, as shown in Figure 6. It is known that the band-gap energy of ZnO decreases with increasing temperature. The change of bandgap energy with temperature is described by Varshni's empirical equation. Assuming that the peak positions of the NBE vary with the temperature

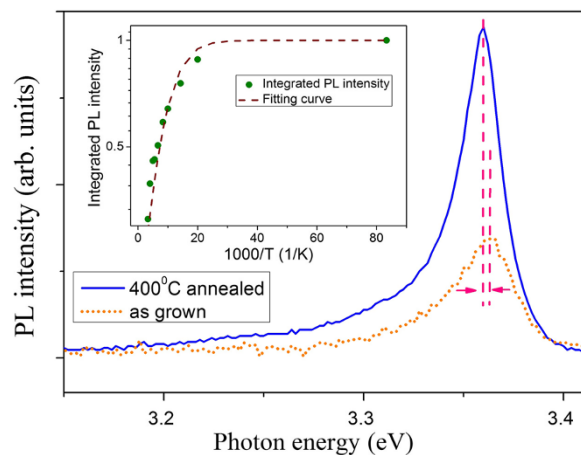


Figure 5 The NBE spectra of the as-grown and annealed ZnO NRs on AZO at 10 K. Inset: the integrated PL intensity of NBE as a function of reciprocal temperature for the annealed ZnO NRs, where the solid circles and the dashed line indicate experimental data and the fitting result to Equation 1, respectively.

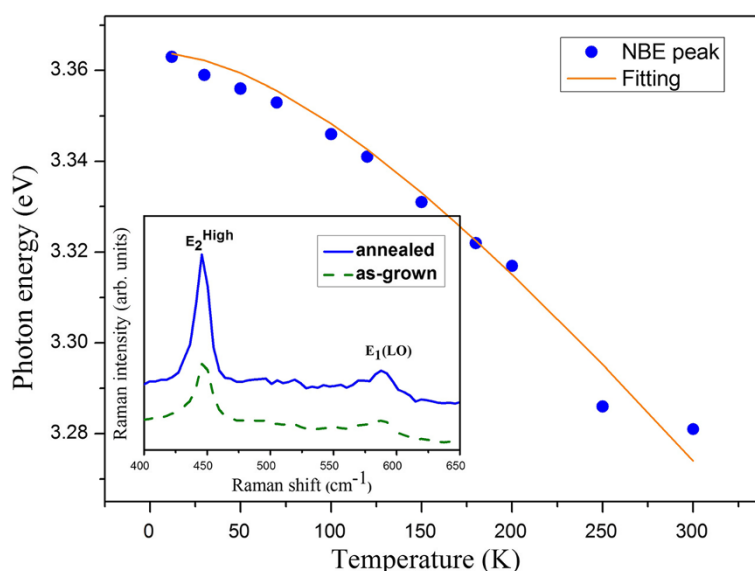


Figure 6 The temperature dependence of NBE emission peak for the annealed ZnO NRs. The solid line represents the fitting result to Equation 2. Inset: backscattering Raman spectra of the as-grown ZnO NRs and ZnO NRs after annealing at 400°C with scattering geometry $z(xx + xy)z$.

as the energy bandgap, the dependence of $E(T)$ on temperature can be fitted with the following expression:

$$E(T) = E(0) - \frac{\alpha T^2}{\beta + T}, \quad (2)$$

where $E(0)$ is the transition energy at zero temperature and α and β are fitting parameters referred to as Varshni's coefficients [20]. Both α and β are material-dependent. The β value is expected to be correlated with the Debye temperature, but a range of values were reported for ZnO [15,20,25]. The fitting results, which are denoted by the solid line in Figure 6, yield $E(0) = 3.364$ eV, $\alpha = 5.5 \times 10^{-4}$ eV/K, and $\beta = 250$ K.

Origin of enhanced optical properties

Based on the results above, we discuss annealing effects on the emission properties of our low-temperature-grown ZnO NRs. The enhancement of NBE by annealing might be explained by two previously proposed mechanisms. One is the elimination of unwanted functional groups acting as nonradiative centers on the surface of ZnO, and the other is the improvement of the crystal quality resulting from removal of intrinsic defects [26]. But the anomalous behavior after 400°C treatment, which shows reduced NBE, cannot be explained by the mechanisms above. In our aqueous solution growth of ZnO, generation of interstitial H is highly possible. Hydrogen defects can be introduced as a result of incomplete dehydration during the formation of ZnO [27,28]. Most of these initial hydrogen states, interstitial H or H complex (hydroxyl group and

complex with other defects), are not active donors. During annealing at 400°C, intrinsic defects are passivated and initially trapped hydrogen is released. The interstitial H has high mobility and can move around the lattice. Since there are oxygen vacancies in the ZnO NRs, the interstitial hydrogen can be trapped inside oxygen vacancies, forming H_O . Substitutional hydrogen at the oxygen site, from many experimental and theoretical investigations, is believed to be an important shallow donor in ZnO [23,28]. The existence of H_O exactly accounts for the observed activation energy in our PL study [23]. The decrease of NBE intensity under prolonged annealing or annealing higher than 400°C can then be attributed to the dissociation of hydrogen donors. Therefore, we attributed our NBE enhancement to the hydrogen donor formation activated by thermal annealing. The inset of Figure 3 shows illustrative diagrams for recombination processes that have taken place in (1) NBE in the as-grown sample and (2) NBE in the annealed sample. From the schematic diagrams, we can understand the redshift of NBE after annealing treatment, which is due to the annealing-induced activation of hydrogen donors.

The visible emission drops exactly in the opposite way as NBE enhances with annealing temperature, that is, it decreases with increasing annealing temperature up to 400°C and then it again increases at 600°C. Figure 3 also shows that visible emission decreases with increasing annealing time as well, but it is noted that after 120 min, further reduction of visible emission is insignificant. The chemical origin of the DE is intriguing. Our result shows that the broad DE is already reduced even with a low-

temperature annealing at 200°C. It is indicative of the presence of OH groups whose desorption temperature is at approximately 150°C [29]. The assignment of this visible emission to the presence of hydroxyl groups is in agreement with a previous report on visible luminescence in ZnO nanocrystals [30]. Moreover, a previous study on the O-H local vibrational modes in ZnO also confirmed that such a low-temperature annealing results in the removal of OH groups [31]. Therefore, we attribute the corresponding reduction of visible (defect-related) emission to the annihilation of OH groups.

Finally, we carried out RT Raman scattering measurements to understand the light-scattering properties. Raman spectra for the nonannealed and 400°C-annealed samples are shown in the inset of Figure 6. Two dominant peaks have been clearly resolved. One peak is observed around 446 cm⁻¹, which is attributed to the nonpolar optical phonon E₂ (high) mode of wurtzite ZnO [21], and the other peak at 588 cm⁻¹, known as E₁ (LO), which is ascribed to the defect formation of oxygen vacancies [32-34]. No other mode related to defect-induced local vibration mode is observed. The weak intensity of E₁ (LO) suggests that there are relatively low oxygen vacancies. Since both spectra contain E₁ (LO) contribution, the low oxygen vacancies should be responsible for the residue visible emission after annealing treatment.

Conclusions

In conclusion, we present the investigation of annealing effect on the optical and structural properties of vertically aligned ZnO NRs on AZO substrates by a single-step aqueous solution method at low temperature. The annealing temperature strongly affects both the NBE and visible emissions. We found the optimum annealing temperature to be 400°C. It yields a sharp and intense NBE emission and effectively suppressed visible emission. The enhancement of NBE is due to the activation of radiative recombinations associated with hydrogen donors while the reduction of DE is ascribed to the annihilation of OH groups. These results are useful to understand and optimize ZnO NRs grown in a low-temperature solution. Our approach has the advantages of low cost, fine quality, and straightforward one-step synthesis, which is promising to realize facile and controlled ZnO-based nanodevice fabrication.

Competing interests

The authors declare that they have no competing interests.

Authors' contributions

DRH directed the project and finalized this manuscript. SEI carried out the sample preparation and XRD, SEM, PL, and Raman measurements. KHS and JJW helped perform the PL measurements. SWK provided equipment support in the synthesis work. CZZ helped prepare the samples. All the authors read and agreed with the final version of the paper.

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